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# Magnetocaloric effect in as-cast $Gd_{1-x}Y_x$ alloys with x = 0.0, 0.1, 0.2, 0.3, 0.4

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In this report, we present the magnetocaloric effect of  $Gd_{1-x}Y_x$  alloys ( $0.0 \le x \le 0.4$ ) prepared by arc-melting from high purity Gd and Y precursors in inert atmosphere. The formation of  $Gd_{1-x}Y_x$  solid solutions was verified by means of X-ray diffraction analysis across the compositional series; also, residual secondary phases Gd and Y were present. Magnetic characterization performed by Vibrating Sample Magnetometry at a maximum applied field of 3.0 T showed a drastic reduction of the magnetization saturation (from 233 emu/g for x = 0.0 to 183 emu/g for x = 0.4), due to a dilution effect of the Y alloying, together with a marked Curie temperature decrease from 296 K to 196 K between x = 0.0 and x = 0.4. The second-order character of the magnetic transition was established by Arrot plots for all the cases. On the other hand, the magnetic entropy variation, determined from numerical integration of Maxwell relation displayed excellent values above 5.30 J/kg K for alloys with x < 0.3 due to the steep transition of the thermomagnetic curves. © 2014 AIP Publishing LLC. [http://dx.doi.org/10.1063/1.4862086]

#### INTRODUCTION

Recently, magnetic refrigeration based on the magnetocaloric effect (MCE) has been a topic of great interest for minimizing environmental impact and global warming.<sup>1–5</sup> A key advantage of magnetic refrigeration is that it does not use chlorofluorocarbons causing ozone layer depletion. Although magnetic cooling to achieve ultralow temperatures dates back to early past century, the interest in its use at near room temperature operation puts the attention on the search for new materials or on the improvement of current ones. Over the past few years there have been many studies of magnetic refrigeration, where ferromagnetic materials, such as Gd, have been used as the magnetic refrigerant.<sup>o</sup> However, since the value of magnetic entropy variation  $\Delta S_M$ of ferromagnetic materials shows a peak near the Curie transition,  $T_C$ , and  $\Delta S_M$  decreases as the temperature separates away from the T<sub>C</sub> temperature, the working temperature region of magnetic refrigeration with ferromagnetic materials has been restricted to short operational temperature ranges. In order to apply room-temperature magnetic refrigeration techniques to practical use, the working temperature region must be greater than 30 K.<sup>7</sup> To solve this problem, it is necessary to increase the working temperature range of magnetocaloric materials by the use of layered structures of magnetic refrigerants with different T<sub>C</sub>. This subject has been the focus in most of the recent work on MCE, where some materials like Gd-Si-Ge based compounds and others, such as (La<sub>1-x</sub>Ca<sub>x</sub>)MnO<sub>3</sub>, MnFeP<sub>1-x</sub>As<sub>x</sub>, and La(Fe,Si)<sub>13</sub>, have attained interesting advances. 4,5,8-11

This work presents a systematic study on the magnetocaloric effect and  $T_C$  dependence on composition for as-cast  $Gd_{1-x}Y_x$  alloys with x = 0.0, 0.1, 0.2, 0.3, and 0.4 as possible candidates for layered-structure magnetic refrigerants.

#### **EXPERIMENTAL**

Raw material ingots of Gd and Y, from Alfa Aesar, with a purity of 99.9%, for both of them, were used to produce small buttons of  $Gd_{1-x}Y_x$ , with x = 0.1, 0.2, 0.3, and 0.4, by arc melting under Ar atmosphere. The buttons were turned and re-melted three times to ensure homogeneity. X-ray diffraction (XRD) patterns were obtained using a Panalytical X'Pert MPD diffractometer with Cu-K $\alpha$  radiation in a  $2\theta$ ranging from  $20^{\circ}$  to  $90^{\circ}$ , with a step size of  $0.016^{\circ}$  and time per step of 30 s. Hysteresis loops and thermomagnetization curves for as-cast alloys were measured on a Quantum Design Versalab system in fields up to 30 kOe at temperature over a temperature interval ranging from 50 to 350 K. The magnetic entropy variation  $\Delta S_M$  was determined by numerical integration of the Maxwell relation based on isothermal magnetization curves measured in a range of temperatures around the T<sub>C</sub>, as

$$\Delta S_M = -\mu_0 \int_{H_1}^{H_2} \frac{\partial M}{\partial T} dH, \qquad (1)$$

where  $\mu_0$  is the vacuum permeability and T is the temperature.<sup>12</sup> The refrigerant capacity (RC), defined as the amount of heat transferred between the hot and cold sinks in an ideal refrigeration cycle,<sup>1</sup> was calculated for all the alloys by numerical integration of the area under the  $-\Delta S_M vsT$ curve.<sup>13,14</sup>

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Material	a (Å)	c (Å)	M <sub>S</sub> (emu/g)	$T_{C}\left(K\right)$	(dM/dT) <sub>max</sub>	$-\Delta S_M(Jkg^{-1}K^{-1})$	RC (Jkg <sup>-1</sup> )
Gd	3.621(2)	5.776(5)	233	296	0.21	5.43	289
Gd <sub>0.9</sub> Y <sub>0.1</sub>	3.636(2)	5.701(4)	231	269	0.29	5.50	284
Gd <sub>0.8</sub> Y <sub>0.2</sub>	3.647(2)	5.725(3)	214	245	0.20	5.35	282
Gd <sub>0.7</sub> Y <sub>0.3</sub>	3.712(3)	5.765(4)	218	218	0.18	4.82	271
$\mathrm{Gd}_{0.6}\mathrm{Y}_{0.4}$	3.665(1)	5.808(3)	183	196	0.09	4.24	232

TABLE I. Cell parameters and magnetic properties of as-cast  $Gd_{1-x}Y_x$  alloys with x = 0.0, 0.1, 0.2, 0.3, and 0.4.

#### **RESULTS AND DISCUSSION**

The formation of  $Gd_{1-x}Y_x$  solid solutions across the compositional series was verified from X-ray diffraction patterns (not shown) indicating hexagonal crystalline structure according to ICDD PDF 01-089-2924, also, secondary phases Gd and Y were present at the minority. Cell parameters for the Gd-Y solid solution were calculated by using UNITCELL software.<sup>15</sup> The obtained values are presented in Table I. It was found that cell parameters were not strongly affected by increment of Y content.

The hysteresis loops measured at 50 K for  $Gd_{1-x}Y_x$ alloys are shown in Figure 1. A marked reduction of the saturation magnetization is manifested for increasing Y content due to a dilution effect caused by the incorporation of non-magnetic Y atoms. A resume of properties is shown in Table I. Thermomagnetic *M* vs *T* curves (H = 100 Oe) for all the alloys within the temperature range from 150 to 350 K are displayed in Figure 2, for which T<sub>C</sub> values were determined at the maximum of the derivative dM/dT. In general, it can be observed a steep transition of the thermomagnetic curves, with only the  $Gd_{0.6}Y_{0.4}$  alloy displaying a smooth transition. The T<sub>C</sub> values were determined as 296, 269, 245, 218, and 196 K for x = 0, 0.1, 0.2, 0.3, and 0.4, respectively. The inset in Figure 2 shows the  $T_C$  as a function of the yttrium content, for which a lineal tendency of T<sub>C</sub> with composition can be observed, characterized by a reduction rate of 2.5K for each 1% of Y concentration. This behavior affords the possibility for a precise control of T<sub>C</sub> in order to establish the temperature range where these Gd-Y alloys could be useful for magnetic refrigeration applications.

Isothermal magnetization curves were measured for all  $Gd_{1-x}Y_x$  alloys in the temperature range of 150 to 300 K. The magnetic field was applied in a very slow sweep rate to ensure an isothermal-like process. Figure 3 shows (a) isothermal magnetization curves and (b) Arrot plots for the  $Gd_{0.7}Y_{0.3}$  alloy, as example. The progressive evolution of these curves to linear behavior denotes a typical ferromagnetic transition in the vicinity of  $T_C$ . The Arrot plots<sup>16</sup> presented in Figure 3(b) shows positive slopes for all isothermal curves, indicating that the Gd-Y alloys undergo a second order Curie transition. Same behavior was recorded for all the remaining alloy samples.

The calculated values for the temperature dependence of  $\Delta S_M$  for all the  $Gd_{1-x}Y_x$  alloys are shown in Figure 4. It is observed that all curves display a caret-like shape indicating that the magnetic phase transition near  $T_C$  is a second-order phase transition,<sup>17</sup> which is consistent with the Arrot plots (Fig. 3).  $\Delta S_M$  values slightly increases initially from x = 0.0 to x = 0.1, followed by a marginal reduction for x = 0.2 and noticeable decrements for x > 0.2, as indicated in Table I. In addition, RC values for the alloys are also exhibited in Table I, for which interesting values between 232 and 289 J/kg are manifested, well comparable with the maximum RC = 289 J/kg corresponding to pure Gd (x = 0.0). The  $\Delta S_M$  behavior can be attributed to the steepness of the *M vs T* plots, quantified by means of the maximum of the d*M*/d*T* 



FIG. 1. Hysteresis loops of  $Gd_{1-x}Y_x$  alloys measured at 50 K with  $H_{max} = 30$  kOe. The saturation magnetization  $M_s$  as a function of the compositional series is shown in the inset.



FIG. 2. Thermomagnetic curves, M vs. T, for  $Gd_{1-x}Y_x$  alloys ( $H_{max} = 100$  Oe). Curie temperature across the compositional series is shown in the inset.

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FIG. 3. (a) Isothermal magnetization curves and (b) Arrot plots for the  $Gd_{0,7}Y_{0,3}$  alloy.



FIG. 4. Magnetic entropy variation ( $\Delta H = 30$  kOe) as a function of temperature for Gd<sub>1-x</sub>Y<sub>x</sub> alloys.

curves. This maximum is included in Table I for each alloy. Those maximums are very similar for x = 0.0, 0.1, and 0.2

(except for x = 0.1, which exhibits an enhanced dM/dT peak), while for x > 0.2, a reducing tendency is manifested (in fact, the lowest slope corresponds to x = 0.4). This behavior is consistent with the observed  $\Delta S_M$  dependence on Y content. Hence, the entropy variation in these Gd-Y alloys seems to be mainly driven by M vs T rate of change around T<sub>C</sub>, rather than the total magnetization variation when the alloys go through the magnetic transition at T<sub>C</sub>.

#### CONCLUSIONS

The magnetocaloric effect of as-cast  $Gd_{1-x}Y_x$  alloys (x = 0.0–0.4) (quantified by means of magnetic entropy variations at  $\Delta H = 30$  kOe) was found very similar for x values up to 0.2, whereas marked reductions were recorded for x > 0.2. The observed entropy variation dependence on composition for these Gd-Y alloys was ascribed to the rate of change of the *M* vs *T* curves around T<sub>C</sub>. On the other hand, the Curie temperature resulted strongly dependent on Y composition, with a linear dependence characterized by a reduction rate of 2.5 K for each 1% of Y added to the alloys. The maximum entropy variation was of -5.5 J/kg K for the Gd<sub>0.9</sub>Y<sub>0.1</sub> alloy with a T<sub>C</sub> = 269 K.

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